

## STUDIES ON GLASS TRANSITION OF LITHIUM-IRON PHOSPHATE GLASSES

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Amorphous analogs of lithium–iron phosphates (LFP) were prepared by standard press-quenching method and their thermal stabilities as well as structural properties were studied for the first time. Glass transition temperature  $T_g$  determined by DTA method was observed at the temperature range 492–523°C, depending on the glass composition. The maxima of crystallization peaks were observed in the 555–579°C range. In products obtained after heating up to 700°C the XRD patterns have revealed the presence of: LiFePO<sub>4</sub> (triphylite), FePO<sub>4</sub> (heterosite),  $\alpha$ -FePO<sub>4</sub> (quartz like structure) and Li<sub>3</sub>Fe<sub>2</sub>(PO<sub>4</sub>)<sub>3</sub> (Nasicon like structure) phases.

**Keywords:** cathode materials, DTA, Li-ion batteries, lithium–iron phosphate glasses, phospho-olivines

### Introduction

Crystalline olivine-like Li<sub>x</sub>FePO<sub>4</sub> solid solutions (also known as LFP – lithium–iron phosphate) are under intensive studies as the most competitive positive electrode materials for Li-ion rechargeable batteries. In these batteries Li<sup>+</sup> cations intercalate from electrolyte into cathode and this process is accompanied with electron flow via external electric circuit. The change of Li content is related neither to change of electro-neutrality nor to change of Fe or P contents. The main advantages of LiFePO<sub>4</sub> as cathode material are that it is a highly stable, cheap and non-toxic material which maintains high theoretical specific capacity of 170 mAh g<sup>-1</sup> and a high discharge voltage [1–6]. Unfortunately, besides all their advantages, olivine-like phases exhibit one serious drawback – very low electrical conductivity – ca. 10<sup>-10</sup> S cm<sup>-1</sup> – at room temperature. Many efforts have been undertaken to improve their electrical properties by structure modifications, introduction of carbon additives [5, 7], or by appropriate doping [8]. Recently we have proposed another, yet unexplored, way of circumventing the problem of low conductivity of crystalline olivine cathode materials. We have prepared vitreous analogs of these materials and have started studies on their long-range and local structure, electrical charge transport and magnetic properties [9]. The second step, being under investigation, consists in turning these glasses into nano-materials by an appropriate thermal treatment.

This work reports our most recent results of studies on thermal and structural properties of LFP

glasses, whose nominal composition can be approximately written as Li<sub>x</sub>FePO<sub>4</sub>, where  $x$  was changed in the 0≤x≤1 range at Δx=0.1 intervals. In particular, we were interested in the determination of the thermal stability range and beginning of crystallization of studied glasses. Our earlier studies of lithium-vanadate phosphate glasses show that nanocrystallization has an especially important effect on electrical conductivity enhancement [10]. The recent results indicate that a similar phenomenon occurs in the LFP glasses [11].

### Experimental

A series of vitreous samples of nominal composition Li<sub>x</sub>FePO<sub>4</sub> for 0≤x≤1 at Δx=0.1 intervals were synthesized by a press-quenching technique. Appropriate amounts of dried precursors: Li<sub>2</sub>CO<sub>3</sub> (Aldrich, 99.99%), NH<sub>4</sub>H<sub>2</sub>PO<sub>4</sub> (POCh – Polish Chemicals, 99.5%) and Fe<sub>2</sub>O<sub>3</sub> (POCh, 98.5%) were ground and mixed in a mortar. Alumina crucibles filled with the powders were placed in an electric furnace and heated from 20 to 1270°C in air at a heating rate of 5 K min<sup>-1</sup>. The molten mixtures kept at 1270°C were rapidly poured out onto a stainless-steel plate held at a temperature close to 25°C and immediately covered by a second stainless-steel plate. The average thickness of the resulting samples was 0.5–1.0 mm.

The composition of the solid products was analyzed by means of X-ray diffraction (XRD) on a Philips X'Pert apparatus using Ni-filtered ( $\lambda=1.54$  Å)

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$\text{CuK}_{\alpha 1}$  radiation in step mode with step size of  $0.01^\circ$ . The pulse count integration time was 0.3 sec.

DTA measurements, in the temperature range of  $20$ – $700^\circ\text{C}$ , were carried out on a DTA7 System thermal analyzer (Perkin-Elmer). The conditions of the experiments were: heating rate  $10 \text{ K min}^{-1}$ , Ar flow rate  $50 \text{ cm}^3 \text{ min}^{-1}$ , sample mass  $130$ – $133 \text{ mg}$ .

## Results and discussion

The DTA curves of the samples reveal the existence of two overlapped thermal effects. The first effect is due to devitrification and the other, strongly exothermic effect is due to the crystallization of phosphate phases in the system (Fig. 1).

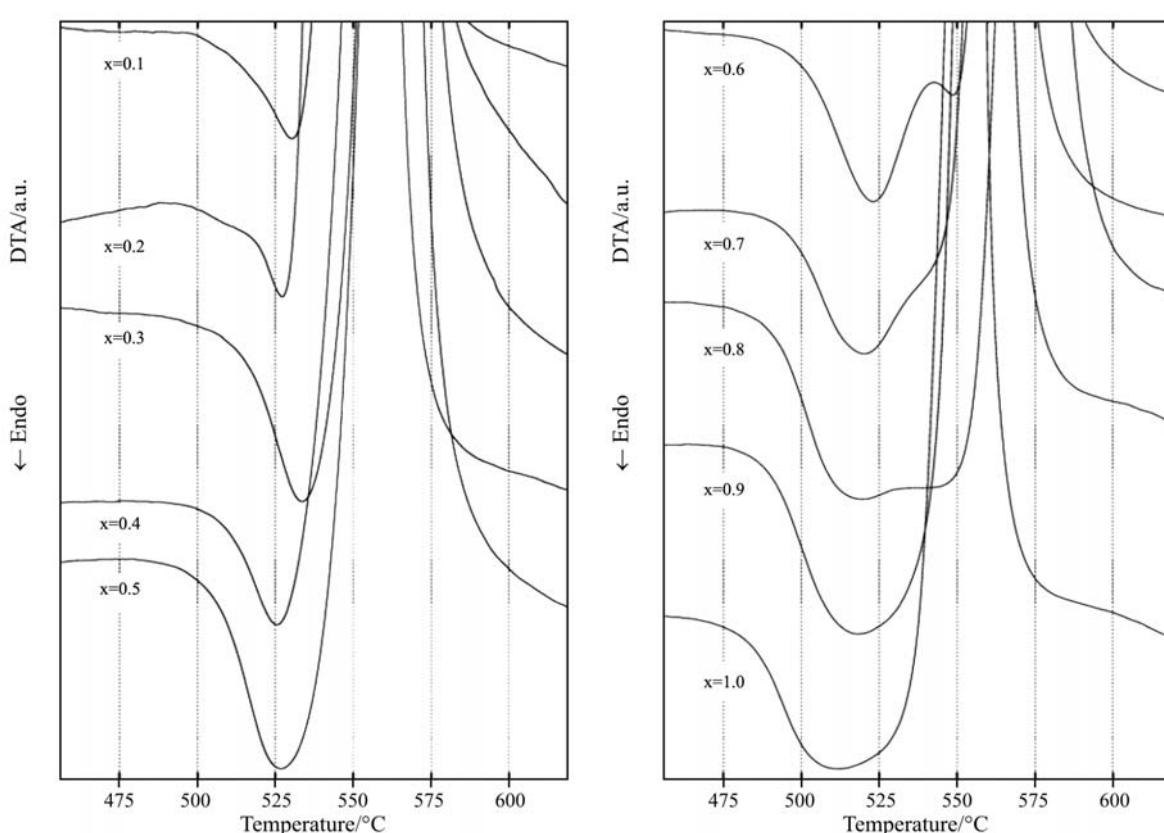
The superposition of the two effects, particularly in the samples with low contents of lithium ( $\text{Li}_{0.1}\text{FePO}_4$  and  $\text{Li}_{0.2}\text{FePO}_4$ ) makes impossible to determine the values of  $\Delta C_p$  during glass transition. In samples containing more lithium the separation of the two effects is clearer; the value of  $\Delta C_p$  determined for the sample of  $\text{Li}_{0.8}\text{FePO}_4$ , in which the two effects are the most readily seen, is  $0.7 \text{ J g}^{-1} \text{ K}^{-1}$ . The sample exhibits the greatest stability of the vitreous phase determined as the  $T_c-T_g$  difference which amounts  $79^\circ\text{C}$  (Fig. 2, Table 1).

Due to the overlapping of thermal effects of glass transition and crystallization mentioned earlier, no relationship between the composition of the studied glasses and the  $T_g$  value was observed. It should be noted, however, that the onset temperature of glass transition decreases with increasing contents of lithium, with the exception of  $\text{FePO}_4$  and  $\text{Li}_{0.1}\text{FePO}_4$  samples (Fig. 3).

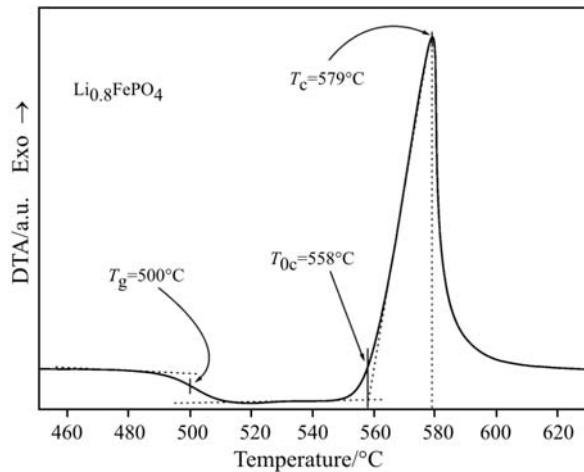
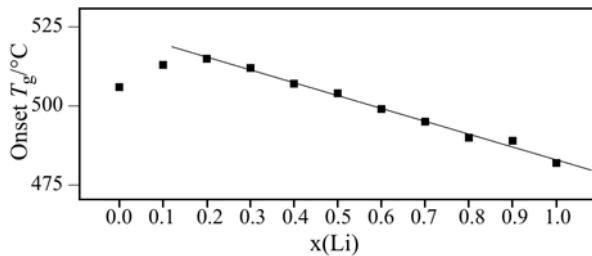
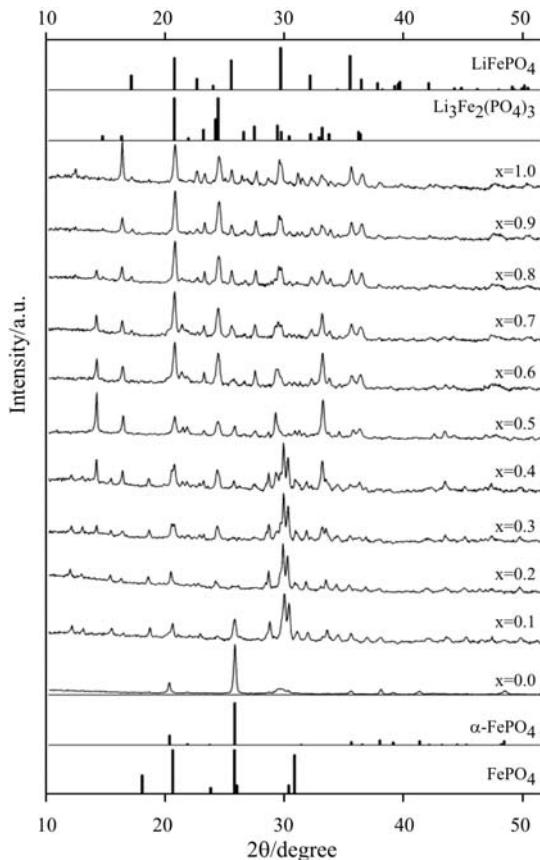
The maximum of the exothermic effect of crystallization of the studied glasses is observed from  $555$

**Table 1** Values of  $T_c-T_g$  vs. lithium content  $x$  for studied glasses

$x$	$T_g/^\circ\text{C}$	$T_c/^\circ\text{C}$	$T_c-T_g/^\circ\text{C}$
0.0	517	574	57
0.1	—	563	—
0.2	—	555	—
0.3	522	568	46
0.4	516	561	45
0.5	514	567	53
0.6	510	571	61
0.7	505	567	62
0.8	500	579	79
0.9	499	563	64
1.0	492	559	67



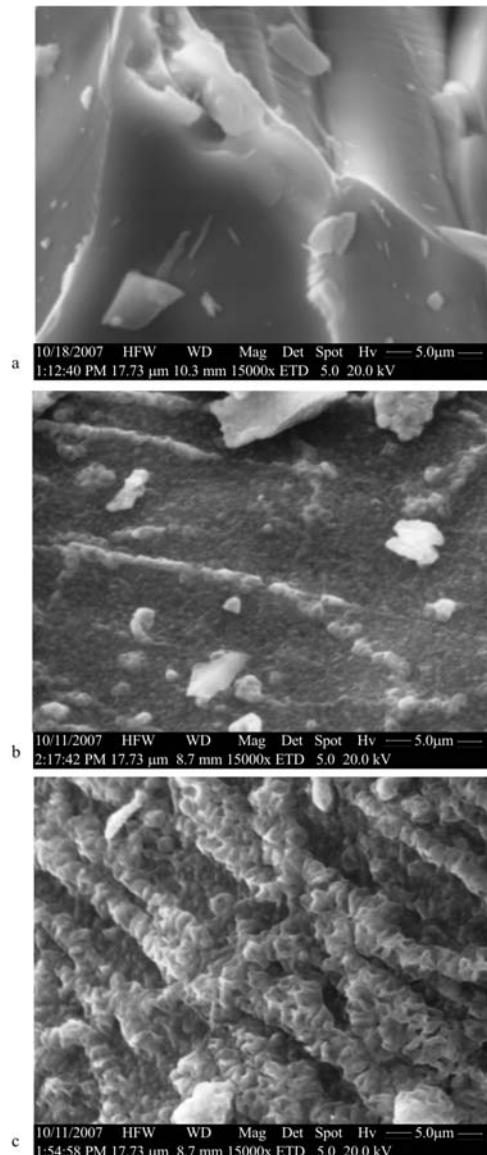
**Fig. 1** DTA curves for  $\text{Li}_x\text{FePO}_4$  glasses

**Fig. 2** DTA curve for  $\text{Li}_{0.8}\text{FePO}_4$ **Fig. 3** Onset temperature of the glass transition of  $\text{Li}_x\text{FePO}_4$  glasses**Fig. 4** XRD curve of studied samples after heating up to  $700^\circ\text{C}$ 

to  $579^\circ\text{C}$ . In the case of samples containing more than 20 mol% lithium the crystallization process can be seen on DTA curves at temperatures of up to  $700^\circ\text{C}$ . In the products obtained at that temperature the XRD measurements have revealed the presence of four phases:  $\text{LiFePO}_4$  (triphylite),  $\text{FePO}_4$  (heterosite),  $\alpha\text{-FePO}_4$  (quartz like structure) and  $\text{Li}_3\text{Fe}_2(\text{PO}_4)_3$  (Nasicon like structure) (Fig. 4).

SEM micrograph of as received glass –  $\text{Li}_{0.8}\text{FePO}_4$  – show smooth uniform area, typical for the amorphous material. After heating up to  $540^\circ\text{C}$  one can see small crystallites, about  $1\ \mu\text{m}$  and much smaller grains around  $100\ \text{nm}$ . Annealing of the sample at  $700^\circ\text{C}$  results in massive crystallization (Fig. 5).

Temperatures  $T_g$ ,  $T_{0c}$  and  $T_c$  determined from DTA curves indicated wide thermal stability range of

**Fig. 5** SEM micrograph of  $\text{Li}_{0.8}\text{FePO}_4$  glass; a – as received, b – after heating up to  $540^\circ\text{C}$ , c – after heating up to  $700^\circ\text{C}$

studied materials and were used to set up the temperature program of impedance spectroscopy measurements [11].

## Conclusions

For the first time, lithium–iron phosphate (LFP) glasses were prepared – the promising cathode materials for Li-ion batteries. The obtained glasses exhibit considerable thermal stability in comparison to other conductive glasses (e.g. lithium–vanadate phosphate glasses [10]). The other advantage of the prepared glasses is their inherent ability to nanocrystallization [11] which may improve their electrical conductivity.

## References

- 1 K. Zaghib, K. Striebel, A. Guerfi, J. Shim, M. Armand and M. Gautier, *Electrochim. Acta*, 50 (2004) 263.
- 2 A. Manthiram and J. B. Goodenough, *J. Solid State Chem.*, 71 (1987) 349.
- 3 A. K. Padhi, K. S. Nanjundaswamy and J. B. Goodenough, *J. Electrochem. Soc.*, 144 (1997) 1188.
- 4 A. B. Bykov, A. P. Chirkov, L. N. Demyanets, S. N. Doronin, E. A. Genkina, A. K. Ivanov-Shits, I. P. Kondratyuk, B. A. Maksimov, O. K. Melnikov, L. N. Muradyan, V. I. Simonov and V. A. Timofeeva, *Solid-State Ionics*, 38 (1990) 31.
- 5 H. Huang, S. C. Yin and L. F. Nazar, *Electrochem. Solid State Lett.*, 4 (2001) A170.
- 6 C. Delacourt, P. Poizot, J. M. Tarascon and C. Masquelier, *Nature Mater.*, 4 (2005) 254.
- 7 Z. Chen and J. R. Dahn, *J. Electrochem. Soc.*, 149 (2002) A1184.
- 8 S. Y. Chung, J. T. Bloking and T. M. Chiang, *Nature Mater.*, 1 (2002) 123.
- 9 A. Ait Salah, P. Jozwiak, K. Zaghib, J. E. Garbarczyk, F. Gendron, A. Mauger and C. M. Julien, *Spectrochim. Acta A*, 65 (2006) 1007.
- 10 J. E. Garbarczyk, P. Jozwiak, M. Wasiucionek and J. L. Nowinski, *J. Power Sources*, 173 (2007) 743.
- 11 P. Jozwiak, J. E. Garbarczyk, M. Wasiucionek, I. Gorzkowska, F. Gendron, A. Mauger and C. M. Julien, *Solid State Ionics*, 179 (2008) 46.

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